# **Supporting Information**

## Cu Electrodeposition on Nanostructured MoS<sub>2</sub> and WS<sub>2</sub> and

## **Implications for HER Active Sites Determination**

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#### Part I: Cu underpotential deposition (UPD)

All Cu UPD tests were performed in Ar sturated 2 mM CuSO<sub>4</sub> in 0.1 M H<sub>2</sub>SO<sub>4</sub> electrolyte. Briefly, the samples were firstly polarized at +0.67 V vs. RHE for 120 s to remove surface oxide species, then copper or hydrogen deposition was carried out at the indicated potential for 100 s. Afterwards, the potential was scanned to +0.67 V vs. RHE (lower potential could not fully strip Cu on MoS<sub>2</sub> and higher potential would cause oxidation of MoS<sub>2</sub>) at a scan rate of 2 mV/s. Background hydrogen adsorption charge  $Q_{BC}$  was calculated by integrating the current obtained upon scanning from the deposition potential to +0.67 V vs. RHE. The Cu stripping charge  $Q_{Cu}$  was calculated in a similar way with correction for background hydrogen adsorption charge.

Potential vs. RHE (V)	$Q_{\rm BC}^{}$ ( $\mu$ C/cm <sup>2</sup> )	$Q_{_{\mathrm{Cu}}}(\mu\mathrm{C/cm^2})$	$Q_{_{\mathrm{Cu}}}/Q_{_{\mathrm{BC}}}$
+0.33	54.6	290.2	5.3
+0.36	42.7	237.7	5.6
+0.39	31.3	166.2	5.3
+0.41	24.0	105.7	4.4
+0.44	16.3	41.1	2.5
+0.47	11.0	9.4	0.8
+0.50	7.5	3.5	0.5
+0.53	5.8	1.4	0.2
+0.55	4.8	0.2	0.1
+0.60	5.3	2.5	0.5

Table S1. Summary of the Cu stripping charge and background hydrogen adsorption charge at different potentials for nanostructured  $MoS_2$ .

Potential vs. RHE (V)	$Q_{\rm BC}^{}$ ( $\mu$ C/cm <sup>2</sup> )	$Q_{_{\mathrm{Cu}}}(\mu\mathrm{C/cm^2})$	$\mathcal{Q}_{_{\mathrm{Cu}}}/\mathcal{Q}_{_{\mathrm{BC}}}$
+0.335	38.9	282.0	7.2
+0.355	35.0	249.1	7.1
+0.385	27.7	217.3	7.8
+0.415	23.7	171.4	7.2
+0.435	20.3	121.6	6.0
+0.455	17.1	59.2	3.5
+0.465	15.7	28.1	1.8
+0.475	14.2	8.6	0.6
+0.485	13.0	4.0	0.3
+0.495	11.8	2.9	0.2
+0.505	10.6	2.2	0.2
+0.535	7.6	0.9	0.1
+0.565	5.3	0.6	0.1
+0.595	3.7	1.1	0.3

Table S2. Summary of the Cu stripping charge and background hydrogen adsorption charge at different potentials for  $WS_2$  film.

### Part II: Comparison between Cu UPD and C<sub>dl</sub>

Sample codes	ALD cycles	Morphology	
WS <sub>2</sub> -1	200	Edge-enriched	
WS <sub>2</sub> -2	200	out-of plane oriented (OoPO)(1)	
WS <sub>2</sub> -3	400	Edge-enriched	
WS <sub>2</sub> -4	400	OoPO	
WS <sub>2</sub> -5	100	Edge-enriched	
WS <sub>2</sub> -6	600	Edge-enriched	
MoS <sub>2</sub> -1	100	OoPO(1, 2)	
MoS <sub>2</sub> -2	400	OoPO	
MoS <sub>2</sub> -3	600	OoPO	
MoS <sub>2</sub> -4	400	Amorphous	

Table S3. Summary of sample codes and corresponding ALD cycles and structures.



**Fig. S1.** Cross-section TEM (a) and HAADF-STEM (b) image of MoS<sub>2</sub> OoPO film; c, d) HAADF-STEM images of edge-enriched WS<sub>2</sub> films. Figures adapted with permission from: a,b ref. 2, Royal Society of Chemistry (2018) and c–d ref. 1, American Chemical Society (2019).



Fig. S2. Linear sweep voltammetry (LSV) curves of different  $WS_2$  films in 0.1 M H<sub>2</sub>SO<sub>4</sub>. Scan rate: 5 mV/s.



Fig. S3. a-f) Cyclic voltammetry (CV) measurements of different WS<sub>2</sub> films.



Fig. S4. Fitting plots showing the extraction of the corresponding  $C_{dl}$  from Fig. S4.



**Fig. S5.** CV tests for corresponding WS<sub>2</sub> films in 0.1 M H<sub>2</sub>SO<sub>4</sub> (black curve) and in 2 mM CuSO<sub>4</sub> in 0.1 M H<sub>2</sub>SO<sub>4</sub> (red curve).



Fig. S6. Linear sweep voltammetry (LSV) curves of different  $MoS_2$  films in 0.1 M H<sub>2</sub>SO<sub>4</sub>. Scan rate: 5 mV/s.



Fig. S7. a-d) Cyclic voltammetry (CV) measurements of different MoS<sub>2</sub> films.



Fig. S8. Fitting plots showing the extraction of the corresponding electrochemical double layer capacitance  $(C_{dl})$  from Fig. S8.



**Fig. S9.** CV tests for corresponding MoS<sub>2</sub> films in 0.1 M H<sub>2</sub>SO<sub>4</sub> (black curve) and in 2 mM CuSO<sub>4</sub> in 0.1 M H<sub>2</sub>SO<sub>4</sub> (red curve).

Part III: X-ray absorption spectroscopy (XAS)



**Fig. S10.** Cu-K edge EXAFS spectra of Cu\_MoS<sub>2</sub>@0.21 V plotted as  $\chi$  (*k*) with *k*-weight of 3. The black curve represents the experimental data and the red curve shows the Feff modeling based on the Feff R space curve fitting result.

	M-4 Model				R sj	pace curve fit
No.	Path	Sub-model	$\mathrm{CN}^{\mathrm{a}}$	R	R	$\pm\sigma^2({\rm \AA}^2)$
1	Cu-S(1)	M-2	2	2.36	2.37	0.0088
2	Cu-Cu	M-3	1	2.56	2.51	0.0053°
3	Cu-S(2)	M-3	3	2.93	2.85	0.0072
4	Cu-S(3)	M-2	2	3.27	3.39	0.0086
5	Cu-Mo(1)	M-2	1	3.46	3.39	0.0098
6	Cu-Mo(2)	M-3	1	3.87	3.95	0.0050 <sup>b</sup>
7	Cu-Mo(3)	M-2	2	4.12	4.13	0.0100 <sup>b</sup>

Table S4. M-4 based R space curve fitting results.

a. Fixed during R space curve fitting guided by DFT model;

b. Limit of parameter floating;





**Fig. S11.** Fit results corresponding to the Cu K-edge EXAFS FT spectra of Cu\_MoS<sub>2</sub>@0.44 V (a) and Cu\_WS<sub>2</sub>@0.465 V (b). Data are plotted as open circles and fits as red line; c, d) Cu-K edge EXAFS spectra of Cu\_MoS<sub>2</sub>@0.44 V (c) and Cu\_WS<sub>2</sub>@0.465 V (d) plotted as  $\chi$  (*k*) with *k*-weight of 3. Black curves represent experimental data and red curves show Feff modeling based on the R space curve fitting.

#### Part IV: Computational Details and Results

The optimized lattice constant is 3.168 Å for the primitive cell, containing one Mo and two S atoms (Mo–S = 2.415 Å and Mo-Mo = 3.168 Å), which is in good agreement with experimental(3) and previous theoretical values.(4) The growth of metal clusters on the basal (0001) plane was modelled on a 5 x 5 supercell of MoS<sub>2</sub> monolayers, which leads to negligible interactions between the system and their mirror images. A vacuum region of 20 Å was added along the normal direction to the MoS<sub>2</sub> monolayers to avoid interactions between adjacent images. The Mo- and S-terminated edges were created by truncating MoS<sub>2</sub> monolayer along the (1010) crystallographic plane, with a vacuum width of 20 Å. The Brillouin zone was sampled using a 9 × 9 × 1 Monkhorst-Pack(5) *k*-point mesh for both the basal plane and the Mo- and S-terminated edges. All structures are relaxed with fixed lattice constants until the Hellmann-Feynman force on each atom becomes smaller than 0.001 eV Å<sup>-1</sup>.

To investigate the possible growth mode and mechanism of Cu and Cu cluster on the MoS<sub>2</sub>, the average binding energy  $(E_{ab})$  was calculated as follows:

$$E_{ab} = \frac{1}{n} \left( E_{total} - E_{subtrate} - n * E_{metal} \right)$$
(1)

where  $E_{total}$  is the total energy of the Cu-MoS<sub>2</sub> system,  $E_{subtrate}$  is the total energy of the MoS<sub>2</sub> substrate,  $E_{metal}$  is the energy of the single Cu atom, and *n* is the number of Cu atoms. According to the above definition, a negative value of  $E_{ab}$  indicates that the process is exothermic and favourable adsorption process. To search the most stable configuration, we considered three high symmetry adsorption sites (**Fig. S15**), *i.e.*, H site (hollow site above the center of hexagons), T<sub>Mo</sub> site (top site directly above a Mo atom), and T<sub>S</sub> site (top site directly above an S atom). The  $E_{ab}$  of a single Cu atom at the H, T<sub>Mo</sub>, and T<sub>S</sub> sites are -1.63, -1.76, and -1.18 eV, respectively, indicating the most stable adsorption site of Cu atoms is the T-Mo, where the Cu atoms form 3-fold coordination with S atoms (**Fig. S16**). Cu dimer and trimer have an  $E_{ab}$  of -1.79, and -2.06 eV, respectively, with the triangular Cu-trimer found to be 0.17 eV more stable than the linearly formed Cu-trimer.

At the Mo- and S-edges, the adsorption of a single Cu atom is found to be energetically more favourable at the Mo-Mo and S-S bridge sites, respectively, as shown in **Fig. S17** and **Fig. S18**. The binding energy of a single Cu atom at the Mo-edge is calculated at -3.33 eV compared to -3.59

eV on the S-edge, suggesting that the S-edge is more active towards Cu adsorption than the Moedge. The stronger binding of the Cu to the MoS<sub>2</sub> edges than to the basal (0001) plane ( $E_b = -1.76$ eV for single Cu atom), suggests that Cu atoms will favourably segregate to the edges rather than grow on the basal (0001) plane.



Fig. S12. Top view of (5 x 5) monolayer of  $MoS_2$  showing the different Cu adsorption sites explored: H=hollow site above the center of hexagons,  $T_{Mo}$ =top site directly above a Mo atom, and  $T_S$ =top site directly above a S atom. Color scheme: Mo = pink and S = yellow.



Fig. S13. Optimized structures and the average binding energies  $(E_{ab})$  of Cu monomer, dimer, and trimer on MoS<sub>2</sub> basal (0001) plane. Color scheme: Mo = pink and S = yellow, Cu = brown.



Fig. S14. Optimized structures and the average binding energies  $(E_{ab})$  of Cu monomer, dimer, and trimer on the Mo-edge of MoS<sub>2</sub>. Color scheme: Mo = pink and S = yellow, Cu = brown.



Fig. S15 Optimized structures and the average binding energies  $(E_{ab})$  of Cu monomer, dimer, and trimer on the S-edge of MoS<sub>2</sub>. Color scheme: Mo = pink and S = yellow, Cu = brown.



Fig. S16. Comparison between convoluted XANES (red) and experimental data (blue) based on 1Cu-bridge-S (a), 2Cu-horizontal (b) and 3Cu-linear (c) model.

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