

# Supplemental Material for: Atomistic Hybrid Particle-Field Molecular Dynamics Combined with Slip-Springs: Restoring Entangled Dynamics to Simulations of Polymer Melts

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## Determining the Number of Slip-Springs from MD Simulations

This section provides more details about the determination of the number of slip-springs in our model. The number of slip-springs  $N_{ssp}$  per chain is expected to depend linearly on the chain length. In practice, we conduct slip-spring hPF-MD simulations with different concentrations of slip-springs  $\rho_{ssp} = N_{ssp}^{\text{total}}/N^{\text{total}}$ , where  $N_{ssp}^{\text{total}}$  is the total number of slip-springs and  $N^{\text{total}}$  is the number of atoms in the system. Then we compute the mean square displacements of the chain centers and extract the values of  $g_1^{\text{mid}}$  at  $t = \tau_R$ . After that, comparing  $g_1^{\text{mid}}(\tau_R)$  of different  $\rho_{ssp}$  with the reference MD values, the optimum total number of slip-springs  $N_{ssp}^{\text{total}}$  for each chain length is identified. In this work, chain length  $N = C_{150}, C_{200}, C_{250}, C_{300}$  and  $C_{350}$  and values of  $\rho_{ssp}$  from 0.03 to 0.08 are chosen for determining the number of slip-springs. An example is shown in Figure S1 for the mean square displacements of chain centers of PE ( $C_{350}$ ) with various  $\rho_{ssp}$  and the reference MD result. It is worth noting that the MD simulation shows the same qualitative behavior as the slip-spring hPF-MD simulations with different  $\rho_{ssp}$ . As seen in Figure S4, the number of slip-springs per chain  $N_{ssp}$  is found to follow the linear relation with the chain length quite well. The linear relation is shown below:

$$N_{ssp} = k N + b \tag{1}$$

where  $k = 0.037$  and  $b = 4.12$  are the fitting parameters. This linearity between the number of slip-springs per chain and the chain length is also found in other slip-spring models<sup>1,2</sup>. With this relation, the number of slip-springs can be extrapolated for long polymers.

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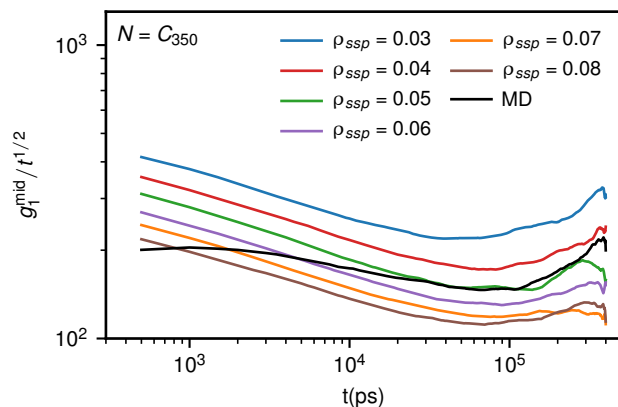


Figure S1: Mean square displacements of chain centers of PE ( $C_{350}$ ) normalized by  $t^{1/2}$  for slip-spring hPF-MD with various concentrations of slip-springs and the corresponding MD simulation.

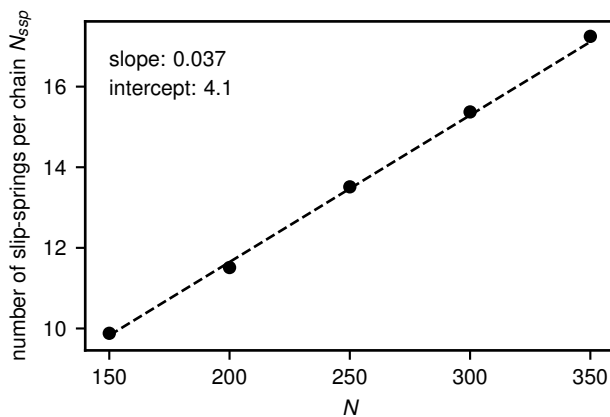


Figure S2: Dependence of the number of slip-springs per chain  $N_{ssp}$  on the polymer chain length  $N$ . The dashed line is a linear fit.

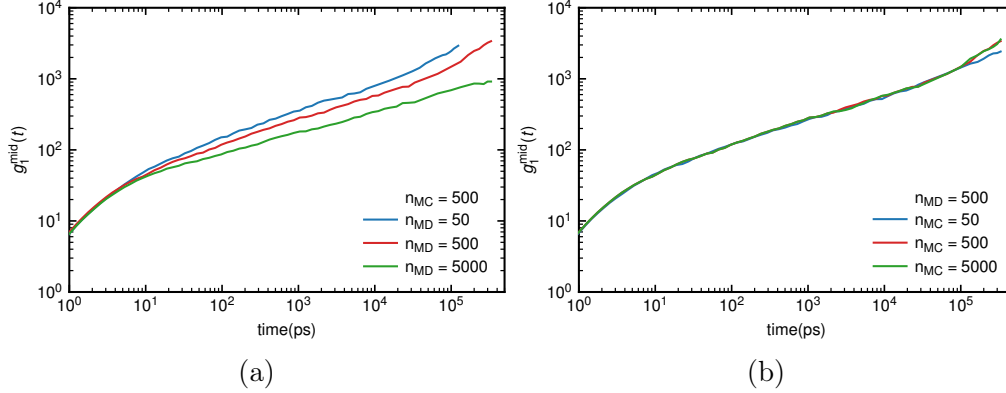


Figure S3: Influence of sequence length of the molecular dynamics and Monte-Carlo blocks on the diffusive dynamics of polyethylene melts composed of 350 carbons.

## Influence of Sequence Length of Molecular Dynamics and Monte-Carlo Blocks

The influence of the sequence length of Molecular Dynamics ( $n_{\text{MD}}$ ) and Monte-Carlo ( $n_{\text{MC}}$ ) blocks on polymer dynamics in the combination of slip-springs and the hybrid particle-field simulation is similar to the initial work of Langeloth et al., which introduces the slip-springs into the dissipative particle dynamics simulation. Basically, the number of MD steps between two MC blocks determines the duration time of the topological constraints (entanglement) imposed by the slip-springs on the polymer chains. Meanwhile, the number of MC steps between two MD blocks dictates the mobility of the slip-springs. In detail, we measure the mean square displacements of the chain center in polyethylene melts of chain length  $N=C_{350}$  to show the impact of different pairs of MD/MC sequence lengths. As seen in Figure 3(a), with the same MC block length  $n_{\text{MC}} = 500$ , the mobility of the polymer melts is profoundly decreasing with increasing MD block length from  $n_{\text{MD}} = 50$  to  $n_{\text{MD}} = 5000$ . On the other hand, the dynamics of polymer melts is expected not to be altered significantly by varying MC block length once it is larger than a critical value at which the slip-springs are mobile enough. Figure 3(b) shows the diffusive dynamics varying as the MC block length from  $n_{\text{MC}} = 50$  to  $n_{\text{MC}} = 5000$ . when  $n_{\text{MC}} = 50$ , the diffusive dynamics is unchanged in the short time regime, while slightly slower than that of both  $n_{\text{MC}} = 500$  and  $n_{\text{MC}} = 5000$  in the long time regime. The overlap of the mean square displacements of  $n_{\text{MC}} = 500$  and  $n_{\text{MC}} = 5000$  indicates that  $n_{\text{MC}} = 500$  sufficiently ensures that the slip-springs are mobile enough in our current systems. To achieve better computational efficiency, we employ  $n_{\text{MC}} = 500$  for all slip-spring hPF-MD simulations.

## Parameters in Slip-Spring Hybrid Particle-Field Simulation

Table 1: Parameters in slip-spring hybrid particle-field simulation

parameters	explanation	units
$T$	temperature of the system	$K$
$\chi_{i,j}$	density-field interaction parameter between particles of type i and j	kJ/mol
$\kappa$	compressibility factor	1/(kJ/mol)
$N$	number of carbons of the polymer chain	-
$M$	number of polymer chains in the system	-
$\rho$	density of the system	kg/m <sup>3</sup>
$K_{ssp}$	slip-spring bond force constant	kJ/mol
$r_{0,ssp}$	equilibrium distance of the slip-spring bond	nm
$N_{ssp}$	number of slip-springs in the system	-
$n_{MD}$	time steps for one molecular dynamics block between Monte-Carlo blocks	-
$n_{MC}$	time steps for one Monte-Carlo block between molecular dynamics blocks	-

## Relationship between the Entanglement Length and Number of Slip-Springs per Chain

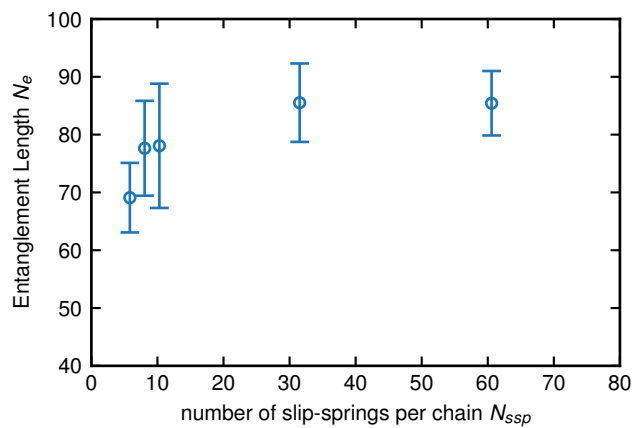


Figure S4: Dependence of the entanglement length  $N_e$  on the number of slip-springs per chain  $N_{ssp}$ .

## References

1. Langeloth, M.; Masubuchi, Y.; Böhm, M. C. and Müller-Plathe, F., *The Journal of Chemical Physics*, 2014, **141**(19), 194904.
2. Sgouros, A. P.; Megariotis, G. and Theodorou, D. N., *Macromolecules*, 2017, **50**(11), 4524–4541.

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